Review Article

Protein crystallography for non-cystallographers
Introduction
Macromolecular crystallography has come a long way in the half-century since the first protein structure (of myoglobin at 6 Å resolution) [1] was published. The establishment of the Protein Data Bank (PDB) [2,3] as the single repository for crystal structures (and later structural models obtained by NMR spectroscopy, fiber diffraction, electron microscopy, and some other techniques) provided a unique resource for the scientific community. The pace of structure determination has accelerated in the last decade due to the introduction of powerful new algorithms and computer programs for diffraction data collection (these days, usually synchrotron-based), structure solution, refinement, and presentation. Of particular importance are structural genomics (SG) efforts conducted in a number of centers worldwide, which can be credited with at least 3500 deposited crystal structures as of September 2007 (W. Minor, unpublished data). Although the total number of protein folds that can be found in nature is still under debate [4] and the structures of many proteins, especially those integral to cell membranes, are still lacking, the gaps in our knowledge are being

Abbreviations
PDB, Protein Data Bank; SG, structural genomics.
Protein crystallography for non-crystallographers

A. Wlodawer et al.

filled quite rapidly. It is now possible to download, with a few clicks of a mouse, the structure of a protein of interest and display it using a variety of graphics programs, freely available to anyone with even the simplest modern computer. Once presented as an elegant picture, the structure seems beyond suspicion as to its validity, or perhaps the validity of its interpretation by its authors. But is that always the case?

An assessment of the quality of macromolecular structures, corrected for technical difficulty, novelty, size, resolution, etc., has recently been published [5]. The authors of that study concluded that, on average, the quality of protein structures has been quite constant over the last 35 years, and there is little difference in quality between structures solved in traditional laboratories and by SG efforts (if anything, the latter are slightly better, at least from some centers). However, a very clear correlation emerged between the quality of the structure and the prestige of the journal in which it was published, with structures in the most exclusive journals being, in general, of statistically lower quality (interestingly, structures published in this journal were found to be, on average, of the highest quality). Of course, the high-impact journals put a proper spin on these results, relating them to the higher complexity of the structures that they accept for publication [6]. However, as interpretation of these structures is at the forefront of structural biology, it is important that readers should be able to assess their quality independently.

The structure of the enzyme frankensteinase (appropriately named after the birthplace of one of the authors of this review, and for some other rather obvious reasons) is presented in Fig. 1A. It certainly looks quite nice, especially to a non-crystallographer, but it does have a few problems, the main one being that no such enzyme exists. However, how could a biochemist or biologist who is not trained in protein crystallography (and, these days, practically nobody is fully trained in this field) recognize this? The purpose of this review is to provide readers with hints that may help them in assessing the level of validity and detail provided by crystal structures (and, to a lesser extent, structures determined by other techniques), define several relevant terms used in crystallographic papers, and give advice on where to find red flags that could affect interpretation of such data. This is not a primer of structural crystallography for non-crystallographers, but rather the musings of four structural biologists, active in various aspects of crystallography, both technical and biological, with a combined total of over 125 years of experience, written for the benefit of those that do not want or need to learn about all the details that go into the solution and refinement of macromolecular structures, but would like to gain confidence in their interpretation.

**How is a crystal structure determined?**

Structural crystallography relies almost exclusively on the scattering of X-rays by the electrons in the molecules constituting the investigated sample. (Some other scattering methods, for example, of neutrons or electrons, although very important, are responsible for only a tiny fraction of the published macromolecular structures.) Because the highly similar structural motifs forming the individual unit cells are repeated throughout the entire volume of a crystal in a periodic fashion, it can be treated as a 3D diffraction grating. As a result, the scattering of X-radiation is enhanced enormously in selected directions and extinguished completely in others. This is governed only by the geometry (size and shape) of the crystal unit cell and the wavelength of the X-rays, which should be in the same range as the interatomic distances (chemical bonds) in molecules. However, the effectiveness of interference of the diffracted rays in each direction, and therefore the intensity of each diffracted ray, depends on the constellation of all atoms within the unit cell. In other words, the crystal structure is encoded in the diffracted X-rays – the shape and symmetry of the cell define the directions of the diffracted beams, and the locations of all atoms in the cell define their intensities. The larger the unit cell, the more diffracted beams (called 'reflections') can be observed. Moreover, the position of each atom in the crystal structure influences the intensities of all reflections and, conversely, the intensity of each individual reflection depends on the positions of all atoms in the unit cell. It is, therefore, not possible to solve only a selected, small part of the crystal structure without modeling the rest of it, in contrast to other structural techniques such as NMR or extended X-ray absorption fine structure which can describe only part of the molecule.

A diffraction experiment involves measuring a large number of reflection intensities. Because crystals have certain symmetry, some reflections are expected to be equivalent and thus have identical intensity. The average number of measurements per individual, symmetrically unique reflection is called redundancy or multiplicity. Because every reflection is measured with a certain degree of error, the higher the redundancy, the more accurate the final estimation of the averaged reflection intensity. The spread of individual intensities of all symmetry-equivalent reflections, contributing to
the same unique reflection, is usually judged by the residual \( R_{\text{merge}} \) (sometimes called \( R_{\text{sym}} \) or \( R_{\text{int}} \)), defined later.

Each reflection is characterized by its amplitude and phase. However, only reflection amplitudes can be obtained from the measured intensities and no direct information about reflection phases is provided by the diffraction experiment. According to the well-established diffraction theory, to obtain the structure of the individual diffracting motif (in our case the distribution of electrons in the asymmetric part of the crystal unit cell), it is necessary to calculate the Fourier transformation of the so-called structure factors, or \( F \) values, which represent the reflection amplitudes and phases. Several methods are used in protein crystallography to determine the phases. Typically, they lead to an initial approximate electron-density distribution in the crystal, which can be improved in an iterative fashion, eventually converging at a faithful structural model of the protein.

The primary result of an X-ray diffraction experiment is a map of electron density within the crystal.
This electron distribution is usually interpreted in (chemical) terms of individual atoms and molecules, but it is important to realize that the molecular model consisting of individual atoms is already an interpretation of the primary result of the diffraction experiment. Finally, the atomic model is ‘refined’ by varying all model parameters to achieve the best agreement between the observed reflection amplitudes \( F_{\text{obs}} \) and those calculated from the model \( F_{\text{calc}} \). This agreement is judged by the residual or crystallographic \( R \)-factor, defined later. It should be stressed that both \( R_{\text{merge}} \) and the \( R \)-factor are global indicators, showing the overall agreement, respectively, between equivalent intensities or observed and calculated amplitudes, and cannot be used to pinpoint individual poorly measured reflections or local incorrectly modeled structural features.

The refinement process usually involves alternating rounds of automated optimization (e.g. according to least-squares or maximum-likelihood algorithms) and manual corrections that improve agreement with the electron-density maps. These corrections are necessary because the automatically refined parameters may get stuck in a (mathematical) local minimum, instead of leading to the global, optimum solution. The model parameters that are optimized by a refinement program include, for each atom, its \( x, y \) and \( z \) coordinates, and a parameter reflecting its ‘mobility’ or smearing in space, known as the \( B \)-factor (or displacement parameter, sometimes referred to as ‘temperature factor’). \( B \)-factors are usually expressed in \( \text{Å}^2 \) and range from \( \sim 2 \) to \( \sim 100 \). [If their values in the PDB files are systematically lower than 1.0, they should be multiplied by 80 \((8\pi^2)\) to be brought to the \( B \) scale.] The \( B \)-factor model used is usually isotropic, i.e. describes only the amplitude of displacement, but more elaborate models describe the individual anisotropic displacement of each atom. Even in the isotropic approximation, crystallographic models of macromolecules are tremendously complex. For example, a protein molecule of 20 kDa would take about 6000 parameters to refine! Frequently, the number of observations (especially at low resolution, vide infra) is not quite sufficient. For this reason, refinement is carried out under the control of stereochemical restraints which guide its progress by incorporating prior knowledge or chemical common sense [7,8]. The most popular libraries of stereochemical restraints (their standard or target values) have been compiled based on small-molecule structures [9–11] but there is growing evidence from high-quality protein models that the nuances of macromolecular structures should also be taken into account [12].

Another way of model refinement, introduced more recently into macromolecular crystallography, involves dividing the whole structure into rigid fragments and expressing their vibrations in terms of the so-called TLS parameters which describe the translational, librational and screw movements of each fragment [13]. Selection of rigid groups should be reasonable, corresponding to individual (sub)domains, for example. An exceedingly large number of very small fragments unreasonably increases the number of refined parameters and leads to models not fully justified by the experimental data.

Although many of the steps in crystal structure analysis have been automated in recent years, the interpretation of some fine features in electron-density maps still requires a significant degree of human skill and experience [14]. A degree of subjectivity is thus inevitable in this process and different people working with the same data may occasionally produce slightly different results. This review is primarily intended to advise those who do not have a deep knowledge of crystallography, but need to know how the objectivity and subjectivity embedded in the available crystal structures should be balanced. Detailed procedures used in macromolecular crystallography are explained in a number of books, some describing them in more advanced terms [15,16], other in simpler ways [17,18].

**Electron-density maps and how to interpret them**

As mentioned earlier, electron-density maps are the primary result of crystallographic experiments, whereas the atomic coordinates reflect only an interpretation of the electron density. Although maps based on the initial experimentally derived phases are sometimes analyzed only by software rather than human eye (a practice that the authors of this review very strongly oppose), we still need to understand what to expect from them.

The basic electron-density map can be calculated numerically by Fourier transformation of the set of observed (experimental) reflection amplitudes \( F_{\text{obs}} \) and their phases. However, because the phases, \( \varphi_{\text{calc}} \), are not available experimentally, they are calculated from the current model. Such a \( (F_{\text{obs}}, \varphi_{\text{calc}}) \) map represents an approximation of the true structure, depending on the accuracy of the calculated phases, that is, on how good the model is from which the phases were computed. Another type of electron-density map, the so-called difference map, calculated using differences between the observed and calculated amplitudes and calculated phases, \( (F_{\text{obs}} - F_{\text{calc}}, \varphi_{\text{calc}}) \), shows the
difference between the true and the currently modeled structures. In such a map, the parts existing in the structure, but not included in the model, should show up in the positive map contours, whereas the parts wrongly introduced into the model and absent in the true structure will be visible in negative contours. In practice, it is customary to use \((2F_{\text{obs}} - F_{\text{calc}}, \varphi_{\text{calc}})\) maps, corresponding to a superposition of both previous maps, to show the model electron density as well as the features requiring corrections. Also, the amplitudes used in map calculation are often weighted by statistical factors, reflecting the estimated accuracy of individual amplitudes and phases.

Because all data used to compute maps (both amplitudes and phases) contain a degree of error, the maps also contain some level of noise. Usually a good display contour for the \((2F_{\text{obs}} - F_{\text{calc}}, \varphi_{\text{calc}})\) map \(~1\sigma\) and for the \((F_{\text{obs}} - F_{\text{calc}}, \varphi_{\text{calc}})\) map about is \(\pm 3\sigma\), where \(\sigma\) is the rmsd of all map points from the average value. Higher contour levels may sometimes be used to accentuate certain features, but the use of lower contour levels may be misleading because this may emphasize noise rather than real features.

It is well established that the appearance of Fourier maps depends more on the phases than on amplitudes. Therefore, even if the correct amplitudes are known from a well-conducted diffraction experiment, inaccurate phases may introduce map bias, which may be difficult to eliminate in the iterative refinement and modeling process. This happens because the wrong phases will always reproduce the same erroneous model features, which in turn will produce the same set of erroneous phases. A map used to overcome such a bias is the so-called ‘omit map’, a variation of the difference map, in which the \(F_{\text{calc}}\) values are computed from a model with the suspicious fragments deleted. Refinement of such a ‘truncated’ model is supposed to remove any ‘memory’ of those fragments in the set of calculated amplitudes and phases. The omit map should then show an unbiased representation of the omitted fragment.

The difference between the initial, experimental and final, optimal electron-density maps is illustrated in Fig. 2. The fragment of the initial map agrees with the final model, but it would not be easy to convincingly build this part of the model into such a map. The map quality is poor because the phases used to construct it were rather inaccurate, and does not result from lack of order, as the protein chain of this fragment is well defined in the crystal, as evidenced by the map calculated with the final phases.

In general, the clarity and interpretability of electron-density maps, even those based on accurate phases, depend on the resolution of the diffraction data (related to the number of reflections used in the calculations). Figure 3 illustrates the appearance of

---

**Fig. 2.** Stereoviews of electron-density maps. The final atomic model of a fragment of the DraD invasin (PDB code 2axw) [79] is superimposed on the maps. (A) The 1.75 Å resolution map calculated with \(F_{\text{obs}}\) amplitudes and initially estimated phases, contoured at the 1.5σ level. This map was used to construct the first model of the protein molecule. (B) The 1.0 Å resolution map calculated with \(F_{\text{obs}}\) amplitudes and the phases obtained upon completion of the refinement, contoured at 1.7σ. The final map shows the complete fragment of the chain with considerably better detail, since it was calculated at much higher resolution (using over five times more reflections) and with very accurate phases.

typical electron-density maps calculated with data truncated at various resolution limits. Whereas at low resolution it is not possible to accurately locate individual atoms, a priori knowledge of the stereochemistry of individual amino acids and peptide groups allows the crystallographer to locate these protein building blocks quite well. With increasing resolution, the maps become clearer, showing separated peaks corresponding to the positions of individual atoms. At atomic resolution, individual peaks are well resolved and their height permits differentiation between atom types. Atomic-resolution maps may show certain non-standard structural features, such as unusual conformations or very short hydrogen bonds. It would not be possible to convincingly model such features into low- or medium-resolution maps. In practice, maps obtained with low-resolution data are even worse than those presented in the Fig. 3, because the relative error of diffraction intensities in the resolution shell of 3.5–3.0 Å for crystals diffracting to 3 Å is much larger than for crystals diffracting to 1.5 Å.

Most proteins contain regions characterized by elevated degree of flexibility. In crystals, such flexibility may result either from static or dynamic disorder. Static disorder results from different conformations adopted by a given structural fragments in different unit cells. Dynamic disorder is the consequence of increased mobility or vibrations of atoms or whole molecular fragments within each individual unit cell. The time scale for such vibrations is much shorter than the duration of the diffraction experiment and, as a result, the electron density corresponds to the averaged distribution of electrons in all unit cells of the crystal. In the case of static disorder, maps are averaged spatially over all unit cells irradiated by the X-rays. In the case of dynamic disorder, the electron density is averaged temporally over the time of data collection. In both cases, the electron density is smeared over multiple conformational states of the disordered fragments of the structure. At low resolution, the smeared electron density may be hidden in the noise and such fragments will not be interpretable, but at higher resolution they may appear as distinct, alternative positions if static disorder is present. Figure 4 illustrates a typical case of a fragment existing in multiple conformations.

A special case of disorder is always present in the solvent region of all macromolecular crystals. The dominating component of the solvent region are water molecules, although obviously any compound...
from the crystallization medium may also be present in the interstices between protein molecules. Some water molecules, hydrogen-bonded to atoms at the protein surface in the first hydration shell, are located at well-ordered, fully occupied sites and can be modeled with confidence. Water molecules at longer distances from the protein surface often occupy alternative, partially filled sites and are difficult to model even at very high resolution. The ‘bulk solvent’ region contains completely disordered molecules and does not show any features except more or less flat level of electron density. This bulk solvent region usually occupies ~ 50% of the crystal volume, although some crystals contain either less or more solvent than usual. The amount of solvent can be estimated from the known protein size and the volume of the crystal unit cell, using the so-called Matthews coefficient [19]. Crystals containing more solvent usually display lower diffraction power and resolution, in keeping with the degree of disorder, which is a consequence of weaker stabilization of the protein molecules through intermolecular interactions.

**A quick look at the files provided by the Protein Data Bank**

Virtually all journals that publish articles describing 3D protein structures require that the authors deposit their results in the PDB. When deposited, each structure is given a unique PDB accession code consisting of four characters. If a structure is later withdrawn or replaced, the code is not reused. Any changes to atomic coordinates result in a new accession code; the code is not reused. Any changes to the experimental details of the structure is also stored in a Crystallographic Information File, 9xyz.cif.) The text file contains a header section with the experimental details and a coordinate section with all experimentally located atoms in the structure of interest. Each atom is identified by an ‘inventory tag’ specifying its name, residue type, chain label, and residue number, which is followed by five numerical values specifying its location (orthogonal x, y, z coordinates expressed in Å), site occupancy factor (a fraction between 0 and 1), and its displacement parameter or B-factor (expressed in Å²), which (at least in theory) provides information about the amplitude of its oscillation. Any person in the world with Internet access can freely download these files or display them on the computer screen using one of several applications available from the PDB site (http://www.rcsb.org/pdb/). For greater flexibility, it is also possible to use one of the more advanced graphical programs, for example, RASMOl [20], PYMOL [21] or COOT [22]. These programs, and some others, provide a variety of ways for displaying and manipulation of the 3D structures and allow their detailed examination.

A file header gives a description of the X-ray experiment, the calculations that have led to structure determination, and some parameters that can help the reader assess the quality of the structure. Traditionally, the ‘Materials and methods’ section of papers that described crystallographic experiments explained in detail how the structure was solved and provided information that allowed the reader to evaluate the quality of the experimental data. Recently, high-impact journals have been enforcing much stricter limits of the size of the papers and, at best, an extract of this information can be found in ‘Supplementary material’ section, which is usually only available online and frequently is not fully reviewed.

Evaluation of structure quality based on the contents of PDB file headers is not easy for non-crystallographers, yet we must stress that any user of such information should look at the header first, before spending too much time looking at the (potentially illusory) details of the structure. A PDB file usually contains information about data extent and quality (resolution, completeness, I/σ, R_merge, both overall and in the highest resolution shell), as well as indicators of the quality of the resulting structure, such as R-factor and R_free (vide infra). In principle, the information that is provided in a PDB deposit should be sufficient to create the ‘Materials and methods’ section by an appropriate software utility. However, the information in the headers of PDB files is often incomplete, contradictory, or erroneous. An extreme case is illustrated by the deposition 2hyd [23] that corrected a series of faulty structures withdrawn from the PDB (together with papers retracted from several high-impact journals, vide infra). The header of the 2hyd.pdb file does not contain any information on how the correct structure was arrived at – all fields that describe structure solution and quality of the data are designated as ‘NULL’. Although, as discussed in the following sections, none of these parameters alone is a rock-solid indicator of the quality of a protein structure, they do provide information that helps in assessing the level of detail that could be gleaned from such a structure. We consider PDB files that do not contain this information to be seriously deficient.
In addition to the text file (e.g. 9xyz.pdb), each crystallographic PDB deposition should be accompanied by a corresponding file with the experimental structure factor amplitudes (9xyz-sf.cif). Most regretfully, for many of the PDB entries no structure factors are available, and even for the most recent depositions (after 1 January 2000) they are found in only 79% of the cases, despite the National Institutes of Health (NIH) requiring that all deposits that have resulted from NIH-sponsored research should include experimental structure factors as well (most other funding agencies have similar rules). The availability of structure factors allows re-refinement of the structure and independent evaluation of model quality and the claimed accuracy of details (although, of course, such checks are not expected to be performed too frequently).

**How to assess the quality of the diffraction data**

The quality of macromolecular crystal structures is ultimately dependent on the quality of the diffraction data used in their determination. The most important indicators of data quality are parameters such as resolution, completeness, $I/\sigma$ (or signal-to-noise ratio), and $R_{merge}$, overall and in the highest resolution shell. It is very important to understand their meaning and the relationship between their numerical values.

**Resolution of diffraction data**

An important parameter to consider when assessing the level of confidence in a macromolecular structure is the resolution of the diffraction data utilized for its solution and refinement (often referred to as resolution of the structure). Resolution is measured in Å and can be defined as the minimum spacing ($d$) of crystal lattice planes that still provide measurable diffraction of X-rays. This term defines the level of detail, or the minimum distance between structural features that can be distinguished in the electron-density maps. The higher the resolution, that is, the smaller the $d$ spacing, the better, because there are more independent reflections available to define the structure. The terms customarily applied to resolution are ‘low’, ‘medium’, ‘high’, and ‘atomic’ (Fig. 5). The appearance of electron density as a function of resolution is shown in Fig. 3. The lowest-resolution crystal structures that have been published with the coordinates start at a resolution of $\sim 6$ Å, which is usually sufficient to provide a very rough idea about the shape of the macromole-

![Fig. 5. Criteria for assessment of the quality of crystallographic models of macromolecular structures. For the resolution and $R$ criteria, the more ‘green’ (i.e. lower) the value, the better. With $R_{free} - R$ and rmsd from ideality the situation is different because there is some optimal value and drastic departures in both directions also set a red flag, although for different reasons. When the difference between $R_{free}$ and $R$ exceeds 7%, it indicates possible over-interpretation of the experimental data. But if it is very low (say below 2%), it strongly suggest that the test data set is not truly ‘free’, for example, because the structure is pseudosymmetric or, even worse, because the test reflections have been compromised in a round of refinement or were not properly transferred from one data set to another. When rmsd(bonds) is very high, it is an obvious signal of model errors. However, when it is very low (e.g. 0.004 Å), it indicates that through too tight restraints the model underwent geometry optimization, rather than refinement driven by the experimental diffraction data. There are different opinions about how rigorous the stereochemical restraints should be. However, because the ‘ideal’ bond lengths themselves suffer from errors in the order of 0.02 Å, it is reasonable to require the model to adhere to them also only at this level.](image-url)
molecule, especially if it contains many helices, as was the case of the first published structure of myoglobin [1]. However, very few crystal structures of even the largest macromolecules are currently published at such low resolution. For example, although early reports of the structure of ribosomal subunits, among the largest asymmetric assemblies studied to date by crystallography, were based on 5 Å data [24], they were quickly followed by a series of structures at 2.4-3.3 Å [25-27]. Today’s standard for medium resolution starts at ~2.7 Å, where there is the first chance to see well-defined water molecules, whose hydrogen-bonding distances are typically that long. Increasingly more structures are now determined to a resolution exceeding 2 Å. The value of 1.5 Å corresponds to typical C–C covalent bonds in macromolecules. When the resolution is significantly beyond this limit (e.g. \( d < 1.4 \) Å), an anisotropic model of atomic displacements can be refined. At 1.2 Å, full atomic resolution is achieved [28,29]. This corresponds to the shortest interatomic distances not involving hydrogen (C=O groups). Direct location of hydrogen atoms in the electron-density map becomes possible at resolution higher than 1.0 Å, because covalent bond distances of hydrogens are in the range 0.9–1.0 Å. The resolution of 0.77 Å corresponds to the physical limit defined by copper \( K\alpha \) X-ray radiation (1.542 Å). Such resolution is very rarely achieved in macromolecular crystallography [30,31], and is beyond the routine limits of even small-molecule crystallography. Ultra-high resolution allows mapping of deformation electron density, for example, of individual atomic or bonding orbitals.

The claimed resolution of a structure determination is sometimes only nominal. If the average ratio of reflection intensity to its estimated error, \( \frac{<I/\sigma(I)>}{I}\), in the highest resolution shell is < 2.0, it can be assumed that the true resolution is not as good. However, if this number is much higher than 2.0, it indicates that the crystal is able to diffract better but the resolution of data was limited by the experimenter or the set-up of the synchrotron experimental station. The use of maximum achievable resolution for refinement not only permits finer structure details to be observed, but also removes possible bias from the model, as higher resolution improves the data-to-parameter ratio.

It has to be noted that the parameters in the PDB deposit header are usually provided for the set of data used for structure refinement, rather than for the data originally used to solve the structure. The set of data used in refinement can be collected with a different experimental protocol than the set of data collected for phasing. For refinement, it is most important to collect a complete data set to the resolution limit of diffraction, whereas for phasing it is most important to collect accurate data at lower resolution, because high-resolution intensities are generally too weak to provide useful phasing signal. For that reason, it is difficult to assess the quality of phasing from the published or deposited information, if a separate experimental data set was used for refinement.

### Quality of the experimental diffraction data

The raw result of a modern diffraction experiment is a set of many diffraction images, stored in computer memory as 2D grids of pixels containing intensities of the individual reflections. The intensities have to be integrated over those pixels that represent individual reflections. Most reflections (together with their symmetry equivalents) are measured many times, and their intensities have to be averaged after the application of all necessary corrections and appropriate scaling. This process is known as ‘scaling and merging’, and its result is a set of unique reflection intensities, each accompanied by a standard uncertainty, or estimate of error. Multiple observations of the same reflection provide a means to identify and reject potential outliers, which may have resulted, for example, from instrumental glitches. However, the number of such rejections should be minimal, a fraction of a percent at most.

As mentioned previously, the accuracy of the averaged intensities can be judged from the spread of the individual measurements of equivalent reflections by the \( R_{merge} \) residual. The simple form of \( R_{merge} = \frac{\sum_h \sum_i |I_{h,i} - \bar{I}_h|}{\sum_h \sum_i I_{h,i}} \) (where \( h \) enumerates the unique reflections and \( i \) their symmetry-equivalent contributors) is not the most useful indicator, because it does not take into account the multiplicity of measurements. More elaborate versions of \( R_{merge} \) have been proposed [32,33], but they are seldom quoted in practice.

A good set of diffraction data should be characterized by an \( R_{merge} \) value < 4–5%, although with well-optimized experimental systems it can be even lower. In our opinion, a value higher than ~10% suggests sub-optimal data quality. At the highest resolution shell, the \( R_{merge} \) can be allowed to reach 30–40% for low-symmetry crystals and up to 60% for high-symmetry crystals, since in the latter case the redundancy is usually higher.

In principle, high multiplicity (or redundancy) of measurements is desirable, as it improves the quality of the resulting merged data set, with respect to both the intensities and their estimated uncertainties. However, in practice this effect may be spoiled by radiation...
damage, initiated in protein crystals by ionizing radiation, especially at the very intense synchrotron beamlines [34,35]. It is not easy in practice to strike an optimal balance between the positive effect of increased multiplicity and the negative influence of radiation damage.

The meaningfulness of measured intensities can be gauged by the average signal-to-noise ratio, \( \langle I/\sigma(I) \rangle \). This measure is not always absolutely valid because it is not trivial to accurately estimate the uncertainties of the measurements [\( \sigma(I) \)]. Usually the diffraction limit is defined at a resolution where the \( \langle I/\sigma(I) \rangle \) value decreases to 2.0.

If the data collection experiment was not conducted properly or if there was rapid decay of diffraction power, some reflections may not be measured at all, and the data may not be 100% complete. Because of the properties of Fourier transforms, each value of the electron-density map is correctly calculated only with the contribution of all reflections, thus lack of completeness will negatively influence the quality and interpretability of the maps computed from such data. Data completeness, that is the coverage of all theoretically possible unique reflections within the measured data set, is therefore another important parameter of data quality.

The above numerical criteria are usually quoted for all data and for the highest resolution shell. Unfortunately, it is not customary to quote these values for the lowest resolution shell, containing the strongest reflections, which are most important for all phasing procedures and for the proper appearance of the electron-density maps. Overall data completeness may reach, for example, 97%, but if the remaining 3% of reflections are all missing from the lowest resolution interval, all crystallographic procedures, from phasing to final model building, will suffer.

As usual, there are exceptions to these rules. This is, for example, the case with viruses, which possess very high internal, non-crystallographic symmetry, in effect increasing the ‘redundancy’ of the structural motif, even if the data may not be complete. For example, for bluetongue virus, 980 individual crystals were used to collect over 21.5 million reflections, and, still the data set was only 53% complete (7.8% in the highest resolution shell). Nevertheless, these data were sufficient for solving the structure [36].

**Structure quality – \( R \), Ramachandran plot, rmsd, and other important \( R \)s**

The quality of a crystal structure (and, indirectly, the expected validity of its interpretation) can be assessed based on a number of indicators. The most important ones will be discussed here in a simplified manner, without any attempt to provide mathematical justification for their use, but only to provide some guidance as to their meaning.

**\( R \)-factor and \( R_{\text{free}} \)**

As mentioned earlier, residuals, or \( R \)-factors, usually expressed as percent, but often as decimal fractions, measure the global relative discrepancy between the experimentally obtained structure factor amplitudes, \( F_{\text{obs}} \), and the calculated structure factor amplitudes, \( F_{\text{calc}} \), obtained from the model. The \( R \)-factor, defined as \( \Sigma|F_{\text{obs}} - F_{\text{calc}}|/\Sigma F_{\text{obs}} \), combines the error inherent in the experimental data and the deviation of the model from reality. With increasingly better diffraction data, frequently characterized by \( R_{\text{merge}} \) of ~ 4% or less, the crystallographic \( R \)-factor is effectively a measure of model errors. Well-refined macromolecular structures are expected to have \( R < 20\% \). When \( R \) approaches 30% (Fig. 5), the structure should be regarded with a high degree of reservation because at least some parts of the model may be incorrect. The best refined macromolecular structures are characterized by \( R \)-factors below 10%. Examples of such structures include xylanase 10A at 1.2 Å resolution [37], rubredoxin at 0.92 Å [38], and antifungal protein EAFP2 at 0.84 Å [39], among others. The atomic resolution structure of L-asparaginase (PDB code 1o7j) describes the positions of over 20 000 independent atoms in the asymmetric unit (including hydrogen atoms), yet it was refined to \( R = 11\% \) at 1 Å resolution [40]. In small-molecule crystallography, where the models contain fewer atoms and the data can be corrected for various systematic errors, it is not unusual to see \( R \)-factors of 1–2%.

An important parameter that was introduced into crystallographic practice in 1992 is free \( R \) [41]. \( R_{\text{free}} \) is calculated analogously to normal \( R \)-factor, but for only ~ 1000 randomly selected reflections (very often inflated to unnecessarily large sets due to blind use of defaults in data reduction software) which have never entered into model refinement, although they might have influenced model definition [42]. In this way, if the mathematical model of the structure becomes unreasonably complex, i.e. includes parameters for which there is no justification in the experimental data, \( R_{\text{free}} \) will not improve (even though the \( R \)-factor may decrease), indicating over-interpretation of the data. This is because the superfluous parameters tend to model the random errors of the working data set, which are not correlated with the errors in the \( R_{\text{free}} \).
set. $R_{\text{free}}$ is an important validation parameter and should set a warning if it exceeds $R$ by more than $\sim 7\%$ (Fig. 5). Its high value may indicate over-fitting of the experimental data, or may result from a serious model defect. For example, addition of an unreasonable number of water molecules into the noisy features of the solvent region will always lower the ordinary $R$-factor, but will not improve $R_{\text{free}}$.

**Modified forms of the $R$-factor**

In addition to the conventional and most popular crystallographic $R$-factor discussed above, other residuals are also in use to gauge the agreement between the real and model worlds. $R_{\text{free}}$ has already been mentioned as a cross-validation parameter based on reflections excluded from refinement. However, its independence from the model is not complete as it may be used to decide on the course of refinement (and model construction). Therefore, an even ‘more independent’ residual, called $R_{\text{sleep}}$, has recently been proposed [42]. That residual should be based on another subset of reflections that are kept in a vault and never used in any calculations, except for the final $R_{\text{sleep}}$ value. Although this concept is methodologically correct, it is not quite certain where to put a limit for sacrifice of the scarce experimental observations on the altar of cross-validation, as removal of consecutive subsets of reflections introduces mathematical errors in the Fourier transformation process (map calculation) and effectively worsens the final map interpretability. A combined application of $R_{\text{free}}$ and $R_{\text{sleep}}$ testing would require 2000–4000 reflections, which might amount to 20% of all observations for a typical data set for a medium-size protein.

Another residual, more common in small-molecule than in protein work, is the weighted $R$-factor or wR2, based on reflection intensities and including the statistical weights with which the observations enter the refinement [43]. The problem of data weighting does not have a good solution in protein crystallography because the uncertainties (errors) estimated for the reflection intensities are not always very reliable. They can be more meaningful if derived from data of high redundancy, i.e. when many observations contribute to the same averaged reflection intensity.

A completely different philosophy is behind the definition of the so-called real-space $R$-factor. Here, the residual is calculated to reflect the correlation between the experimental electron-density map and the one generated purely from the model. Real-space $R$-factors are used less frequently; the disadvantage is that even the experimental map is, in most cases, based on model-derived phases. An important advantage is that map $R$-factors can be calculated selectively for different regions of the model, thus easily revealing the troubling parts, something that is not obvious from the diffraction-space residuals.

**Root-mean-square deviations from stereochemical standards**

Rmsd from standard stereochemistry indicate how much the model departs from geometrical parameters that are considered typical, or represent chemical common sense based on previous experience. Usually the same standards are used as restraints (with adjustable weights) during structure refinement [9,10]. Different parameters can be evaluated by the rmsd criterion, but it is most common to use the value for bond lengths when comparing different models. Good-quality, medium-to-high-resolution structures are expected to have a rmsd(bond) of $\sim 0.02\AA$ (Fig. 5), although numbers half that size are also acceptable. When this number becomes too high ($> 0.03\AA$), it signifies that something might be wrong with the model. It is not desirable to lower this value at all costs, because the standards represent some averages and are themselves not error-free [12]. At very high resolution, the restraint control of model geometry (at least in well-defined areas) becomes less important because the experimental information strongly determines the course of the refinement.

**Ramachandran plots and peptide planarity**

The global deviations of stereochemical parameters from their expected values, discussed above, might raise questions about the quality of the structure but would not pinpoint the source of possible errors. To trace them, one normally runs a geometry validation program, such as PROCHECK [44] or MOLPROBITY [45], to look for indications of curious features. A particularly useful tool is the Ramachandran plot [46], showing the mapping of pairs of $\phi/\psi$ torsion angles of the polypeptide backbone (defined in Fig. 6) against the expected contours. The $\phi/\psi$ angles have a strong validation power because their values are usually not restrained in the refinement (unless a special torsion-angle-refinement method is used) [47]. Two examples of Ramachandran plot are shown in Fig. 7. For the *Erwinia chrysanthemi* L-asparaginase structure (PDB code 1o7j; Fig. 7A), $> 90\%$ of the angles are found in the most favored region of the diagram. One residue, Thr204, is found in the disallowed region, but its strained conformation was well documented in that
and other asparaginase structures [40], thus this departure from ideality can be accepted with confidence. That is not the case with the Ramachandran plot (Fig. 7B) for the structure of the C3b complement pathway protein (PDB code 2hr0), which appears to suffer from a multitude of problems (vide infra).

The third main-chain conformational parameter, the peptide torsion angle $\omega$, is expected to be close to $180^\circ$ or exceptionally to $0^\circ$ for cis-peptides (the latter situation may be more frequent than originally thought). The peptide planes are usually under very tight stereochemical restraints, although there is growing evidence that deviations of $\pm 20^\circ$ from strict planarity should be treated as not abnormal [12,38,48]). Unreasonably tight peptide planarity restraints may lead to artificial distortions of the neighboring $\phi/\psi$ angles in the Ramachandran plot. However, sometimes one encounters in the PDB protein structures with totally impossible peptide-bond torsion angles. Models containing such violations should be regarded as highly suspicious.

Can we trust the published macromolecular structures?

In our opinion, the general answer to this question is a definite ‘yes’, although, as shown below, some problems may be encountered in individual cases. We
Misrepresentation of crystallographic experiments

Fortunately for the field, known cases of outright fabrication of crystallographic data are extremely rare, maybe because the technique is so heavily based on calculations that data are not easy to fake. Perhaps the best known case of that sort was a discovery that the published diffraction patterns attributed to valyl tRNA were actually those of human carbonic anhydrase B [49]. That substitution was detected by analyzing the unit cell parameters of the published diffraction photographs – their values are quite characteristic for a given crystal, although they might bear chance similarity to crystals of other macromolecules. In that case, the latter possibility was ruled out through careful analysis of other aspects of the presented data.

A case of possible manipulation of diffraction data has recently been described (but it must be stressed that, as of the time of writing of this review, it is not yet officially proven). It was pointed out that the data deposited in the PDB for the structure of protein C3b in the complement pathway, refined at 2.26 Å resolution (PDB code 2hr0), are inconsistent with the known physical properties of macromolecular structures and their diffraction data [50]. For example, the deposited structure factors did not show any indication of the presence of bulk solvent, the electron density of the presumably largely unfolded domain was excellent, and there was no correlation between surface accessibility and the atomic B-factors. In addition, some other features (18 distances between non-bonded atoms of < 2 Å, several peptide torsion angles deviating from planarity by as much as 57°, and 4.2% of outliers in the Ramachandran plot, almost all in one subunit; Fig. 7B) are clear indications of serious problems with this structure.

Honest errors in structure determination

In our experience, serious errors in describing a whole macromolecule are rare, especially nowadays, although errors in some local areas might be more common. A structure of ribulose-1,5-biphosphate carboxylase-oxygenase with the chain of one of the subunits traced completely backwards was published [51], but, in a way that should reassure non-crystallographers, the error was noted almost immediately [52]. The statement found in the abstract of the latter publication ‘one of these models is clearly wrong’, paraphrasing the way Winnie-the-Pooh was addressed by Rabbit (‘one of us was eating too much, and I knew it wasn’t me’) [53], is an excellent indication of the self-correcting potential of the collective experience of the crystallographic community. A later re-enactment of this case [8] showed that, although it is possible to refine a backwards-traced structure at medium resolution to acceptable values of R and rmsd(bond), the value of R_{free} would remain completely unacceptable (in that case, 61.7%), clearly indicating that the model was in error. With the mandatory use of R_{free}, similar errors are unlikely to happen again.

A very recent case of an important series of structures that were seriously misinterpreted points out the danger introduced by deviation from standard crystallographic procedures and by over-interpretation of low-resolution data. The structure of the MsbA ABC transporter protein [54], as well as several related structures published by the same group, had to be retracted after the structure of Sav1866, another member of the family, was published [23]. All structures of these very important integral membrane proteins were solved at low resolution. The structure of MsbA was refined using non-standard protocols that utilized multiple molecular models, and this approach may have masked problems that would have been obvious had the authors stayed with more traditional refinement techniques. It must be stressed that all these structures were very difficult to solve and even the apparently correct structure of Sav1866 is characterized by rather high values of R and R_{free} (25.5% and 27.2%, respectively), although such values are not unusual at 3 Å resolution.

Unlike the very rare cases mentioned above in which the whole structures were questionable, local mis-tracing of elements of the protein chain has been more common. A number of such cases have been reviewed previously [8]. Although this type of error may matter very little if it happens to be limited to an area of the protein that is remote from the active site or from site(s) of interaction with other proteins, in other cases it may lead to misinterpretation of biological processes. One well-known case, in which modeling a β strand instead of a helix led to postulating a doubtful model of autolysis, was provided by HIV-1 protease [55]. However, similar to the cases mentioned above, the implausibility of the original interpretation became clear almost immediately, when, first, the structure of a related Rous sarcoma virus protease became available [56], and, soon thereafter, when the
structure of HIV-1 protease itself was independently determined [57].

One important practical aspect of crystallographic structures is to provide details of the interactions between macromolecules (usually enzymes) and small- or large-molecule inhibitors. Interpretation of such structures depends very much on the quality of the electron density for the inhibitor. In some cases, such as the complex of botulinum neurotoxin type B protease with a small inhibitor BABIM [58], the structural conclusions had to be later retracted, although the crystallographic quality indicators appeared to be more than acceptable (resolution 2.8 Å, \( R = 16.2\% \), \( R_{\text{free}} = 23.8\% \)). Similarly, the validity of the structure of a complex of the same enzyme with a target peptide was questioned [59], because the 38-residue peptide was apparently fitted to a very noisy map that could not support the interpretation of its structure.

**Interpretation (and over-interpretation) of structural models**

Assuming that the reader has looked at the header of the PDB file and become convinced that there are no indications of any problems with the diffraction data or with the results of the refinement, what other properties of the structure should be considered? An important aspect of macromolecular crystal structures is the description of solvent areas, as water plays a vital role in the structure of biomolecules and often influences protein function. Another important aspect of the structure is the description of other ligands, especially bound metals. Subsequent interpretation of the structures in terms of known biological and biochemical properties is a crucial step in structural biology. It is also necessary to consider whether the features described in the PDB deposit, such as, for example, placement of hydrogen atoms, could be justified by the resolution and quality of the experimental data.

**Solvent structure**

The solvent content of protein crystals was first analyzed by Matthews [19] on the basis of the few protein crystal structures known at that time, and was found to range from 27 to 65%. Examination of the current contents of the PDB indicates that this estimate is still valid, with an average of 51%, although some exceptions are present. However, the apparent solvent content of entries such as 2avy (92%) or 1q9i (2.0%) certainly indicates errors in the PDB. The presence of such errors (10 cases with solvent content below 2.5%) must be recognized by the users of this database.

Because X-ray crystallography can observe only objects that are repeated throughout the entire volume of the crystal in a periodic fashion, only well-ordered solvent molecules can be identified in the electron-density map. Moreover, the number of observed water molecules also depends on the resolution of the experimental data. To get a rough estimate of the expected ratio of the number of water molecules to protein residues one should subtract the resolution (in Å) from 3. This indicator could be higher (by up to 100%) for crystal structures with a high solvent content (Matthews coefficient > 3.0 Å³Da⁻¹). Thus at low resolution (~ 2.5 Å) it should be possible to identify in the electron-density maps at most 0.3–0.5 ordered water molecules per protein residue and at very high resolution (1.0 Å) this may increase to 2 water molecules per residue. Structures exceeding these limits may contain errors.

It should be noted that the inclusion of a water molecule in the model usually increases the number of refinement parameters by four (three coordinates plus the isotropic B-factor) and subsequently decreases the \( R \)-factor, so assigning water to each unidentified section of density is very tempting, but may not be justified. The presence of water molecules with high B-factors (> 100 Å²) indicates that the solvent structure was not refined very carefully. A large difference in the values of the B-factors for a solvent molecule and its environment is also very suspicious.

**Metal cations**

Around 30% of all PDB deposits report the presence of ordered metal ions, with ~ 20% containing a metal located in a site important for the biological activity of the macromolecule. Functional analysis of a number of proteins crucially depends on the ability to identify possible metal ions in an unambiguous way. Unfortunately, PDB files do not contain any information about the procedures that were used for metal assignment and refinement, and even the relevant papers often relegate this information to supplements. Sometimes metal positions are determined directly, utilizing their anomalous scattering of X-rays. Application of this procedure provides the highest credibility, but most often the metals are assigned simply to the high peaks of electron-density maps. When assigning metal ions in the latter way, the experimenter should have examined the number of ligands, the geometry of the coordination sphere, and the B-factor of the ion and its environment. For example, the distance between calcium and oxygen atoms should be ~ 2.40 Å and between magnesium and oxygen ~ 2.07 Å [60]. If the
distances between a putative calcium and the neighboring oxygen atoms are around 2.1 Å, two possibilities should be considered: (a) a magnesium ion is present, but the experimenter has wrongly assigned the density to calcium; or (b) the refinement was performed with inappropriate restraints. Metal ion distance restraints are necessary especially for lower resolution data, where the observation-to-parameter ratios are usually insufficient for unrestrained refinement [12]. Certain metals have preferences for a particular type of coordination, for example Mg$^{2+}$ tends to show octahedral coordination, whereas Zn$^{2+}$ is most often tetrahedral [60–62]. A useful tool for differentiating between various metal ions is the bond valence concept, which takes into account the valence of the metal and the chemical nature of the ligands [63–65]. An example of an ion assigned as Mg$^{2+}$ that violates most of the rules given above is shown in Fig. 1B. Unfortunately, this part of the structure of frankensteinase was copied directly from the file 1q9q deposited in the PDB.

Whereas the presence in a structure of a few metal ions with acceptable distances to the protein and good geometry should be considered normal, the presence of too many such ions that do not make reasonable contacts with the protein should be a matter of concern. For example, the 2.6 Å structure of *Thermus thermophilus* RNA polymerase (PDB code 1iw7) contains 485 Mg$^{2+}$ ions, the vast majority far beyond 2.07 Å from the nearest oxygen atom. We may safely assume that the identity of most of these ions is very dubious, to say the least.

**Placement of hydrogen atoms**

Hydrogen atoms lack the electronic core and, in molecules of chemical compounds, their single electron is always involved in the formation of bonds. Hydrogen atoms are therefore the weakest scatterers of X-rays, and even in small-molecule crystallography their direct localization is difficult. The only chance to directly localize them in macromolecular structures is in the difference map after the rest of the structural model has been carefully refined at very high resolution. However, even for those proteins that diffract X-rays to ultra-high resolution, only a fraction of all hydrogen atoms can be identified in such maps.

Although hydrogen atoms are not easy to localize directly, they are obviously present in all proteins, sugars, and nucleic acids, and are involved in many biological processes. The location of most of them can be calculated with good accuracy from the positions of the heavier atoms. As a consequence, it is advisable to include the majority of hydrogen atoms in a structural model at calculated positions, and refine them as 'riding' on their parent atoms. In this way, their parameters are not refined independently, but their coordinates are recalculated after each refinement cycle and their contribution to X-ray scattering is correctly taken into account. Some refinement programs have options for such a treatment of hydrogen atoms in an automatic way. At high resolution their contribution may result in a drop of the overall $R$-factor by a few percent. Moreover, if H atoms are not included, their contribution is represented completely by the parent atom and its position tends to refine to the 'center of gravity' of both atoms. As a result the geometry of the refined model may be slightly distorted.

Unfortunately, whereas this method is applicable to most hydrogen atoms, which are rigidly connected within such groups as methylene, amide, phenyl, etc., some other hydrogen atoms, often the most interesting from the chemical and biological point of view, e.g. those within hydroxyl groups or within functions that can be easily (de)protonated, such as carboxyl or amino groups, cannot be treated in this way. In some cases, when the model is accurate enough and refined at high resolution, their presence can be inferred indirectly by analyzing the geometry of the chemical environment (Fig. 8A). For example, if the two C–O bond lengths within a carboxyl group differ significantly, then most probably this acidic group is not ionized. The internal C–N–C bond angles in heterocyclic rings, such as in the imidazole ring of histidine, tend to be by up to 5° wider if the nitrogen atom is protonated [66]. In structures refined at ultra-high resolution, as well as in structures obtained by neutron diffraction (a technique not discussed here, but whose utility is well documented) [67,68], positions of some hydrogen atoms can be visualized directly (Fig. 8B).

Some low-resolution coordinate sets were deposited in the PDB with hydrogen atoms that were utilized during the refinement, but which clearly cannot have any experimental basis in structures solved at low resolution. Some examples are provided by 1pma (3.4 Å), 1gtp (3.0 Å), 1pfx (3.0 Å), or 1ned (3.8 Å), among others. The reader might safely assume that these hydrogen atoms were only modeled and not determined experimentally.

**Catalytic mechanism**

The crystal structure of a nucleic acid complex of the enzyme onconase [69] may represent a case in which the interpretation of the structural results contradicts the established picture by going beyond what can be justified by the extent and quality of the diffraction
data. The authors postulated a novel catalytic mechanism involving the attack on the phosphodiester bond by the His2 imidazole atom of the crucial catalytic His97 residue rather than by Nε1, as is the case with other RNase A-like enzymes. The orientation of the His97 ring in the deposited structure (PDB code 2i5s) was determined, on the basis of the B-factors of the imidazole atoms, to be opposite to that found in all other related structures. However, the interpretation may be an example of trusting crystallographic data beyond the level of credibility. First, the 1.9 Å diffraction data were not of the highest quality (Rmerge = 12.5%). Second, the final refined model places the 'catalytic' nitrogen 4.15 Å from the atom being attacked, at an angle that prevents the creation of any hydrogen bonds. It seems to us more likely that either the side chain of His97 might have been trapped in a non-productive orientation, or the refined values of the B-factors, and in consequence the deduced orientation of the histidine ring, were influenced by data errors.

Is the structure relevant to explanation of the biological properties?

Infrequently, a macromolecular structure may be completely correct in crystallographic terms, yet the coordinates may not correspond to the biologically relevant state of the molecule. A few examples illustrate this situation. The first structure of the core domain of HIV-1 integrase (PDB code 1itg) contained a cacodylate molecule derived from the crystallization buffer attached to a cysteine side chain located in the active-site area [70]. This led the constellation of the catalytic residues Asp64, Asp116, and Glu152 to assume a non-native configuration, although the distortion of the catalytic apparatus became apparent only later, by comparison with other, unperturbed structures, notably the catalytic domain of integrase from avian sarcoma virus [71,72]. The most significant consequence of the inactive conformation of the catalytic residues was the inability of the two aspartate side chains to bind a catalytic divalent metal cation in a coordinated fashion. Subsequent studies of Mg2+ complexes of HIV-1 integrase crystallized in the absence of cacodylate were in full agreement with the structures of other related enzymes [73,74].

A different example of the difficulties in gaining mechanistic insights from high-resolution structures of enzymes is provided by a comparison of crystal structures of the proteolytic domain of Lon proteases belonging to two closely related families, A and B. Structural and biochemical investigation of such a domain of Escherichia coli Lon A (EcLonA; PDB code 1rre) [75] established the presence of a catalytic dyad consisting of Ser679 and Lys722. However, the subsequently determined structure of a corresponding domain of Methanococcus jannaschii Lon B (MjLonB; PDB code 1xhk) indicated the presence of a catalytic triad, which, in addition to the two residues equivalent to the ones mentioned above, also included Asp675 (E. coli numbering) [76]. Such an important structural difference was interpreted in terms of a different catalytic mechanism for these closely related enzyme families. However, atomic-resolution crystal structure of the catalytic domain of Archaeoglobus fulgidus Lon B (AfLonB; PDB code 1z0w) [77], as well as high-resolution structures of a series of mutants, established a
different picture, in which the strand including Ser679 was turned towards solvent, disrupting the catalytic dyad. Mutation of Asp675 to Ala did not affect the activity of the enzyme. The final conclusion, possible only because of the availability of a whole series of structures, was that in the absence of a substrate, product, or inhibitor, the catalytic domain of Lon may adopt an inactive conformation. This is a lesson worth remembering.

A practical approach to evaluating protein structures

Having presented a brief outline of the process leading to the solution of crystal structures and after discussion of the appearance of the electron-density maps and the indicators of quality of both the experimental data and the resulting structures, it is time to summarize some practical approaches to the evaluation of macromolecular structures presented in the scientific literature. A nice picture showing a rendered tracing of the main chain and a few side chains may convey an impression that the structure should be interpreted as is, and even frankensteinase (Fig. 1) may appear to represent at this level a properly solved protein structure. What are the most important indicators that one should pay attention to?

The first thing to check is whether the level of detail of a published structure and the biological inferences drawn from it are justified by the data resolution. One simple indicator to check is the number of modeled water molecules per residue. For example, one of the 1.9 Å structures of a small peptide refined by water molecules per residue. For example, one of the simple indicators to check is the number of modeled

Another resolution-related problem is whether individual B-factors were refined, or whether only an overall B (for the whole structure) or group B-factors (for each residue) were refined. Any structure at resolution lower than ~3 Å in which B-factors were refined individually for each atom should be taken with a grain of salt, because the procedure introduced too many parameters. The structure 1q1p mentioned above is an example of such an approach (refinement of too many parameters and addition of too many solvent molecules often go together). Another example of refinement that is subject to both reservations is provided by a structure also discussed previously, namely DNA polymerase β (PDB code 1zqr) refined at 3.7 Å, in which the final model contains 326 protein residues, 146 water molecules, 3 metal ions, and 15 nucleotides, all refined with individual B-factors that range from 1.0 to 100.0 Å². For high-resolution structures, anisotropic representation of atomic displacement parameters is substantiated only if the resolution is better than ~1.4 Å. At lower resolutions the use of six refined anisotropic parameters instead of one isotropic B-factor is not warranted by the number of reflections available for refinement. Thus a structure of mistletoe lectin I (PDB code 1onk), refined anisotropically at 2.1 Å resolution, is a good example of a procedure that should better be avoided.

The next parameters to consult would be the other two ‘Rs’ presented in Fig. 3. In typical situations, the three criteria should be congruous, i.e. high-resolution structures are expected to be characterized by lower R-factors and better geometrical quality. However, these parameters should not be in the alarming red regions. An example, the structure of eye-lens aquaporin (PDB code 2c32), refined with individual atomic B-factors using data extending only to 7.01 Å resolution, with R = 39.0% and R_free = 38.7%, seems to be unacceptable for several of the reasons given above. The 2.2 Å structure of ferric binding protein (PDB code 1d9y) is characterized by R = 18.5% and R_free = 37.7%, with very large differences between the B-factors of neighboring atoms and groups, with anisotropically refined Fe and S atoms, and with no record of geometry indicators such as rmsd(bond) given in the PDB file. This structure and ones like it should also raise significant concerns.

To further evaluate a structural model, the reader may use validation programs such as PROCHECK [44] or MOLPROBITY [45]. Presumably they have already been used by the authors, but the results are not always summarized in the articles. We especially recommend checking the Ramachandran plot to make sure that no
unexplained torsion angles are found in disallowed regions. Although no longer commonly found in publications, such a plot is available on the PDB website for each deposited structure. It may be a good habit to have a look at the atomic coordinate section of the PDB file to see the level of the B-factors of the atoms that are in the most important fragments of the structure. If one sees values systematically > 40 Å², the fragment may not be well defined at all. An even more important parameter to check is the occupancy factor. Well-ordered atoms should have 1.00 in this column of the PDB file. Values equal to 0.00 mean that such atoms are completely fictitious, without any support from the experiment, added only to mark the chemical composition of the protein sequence. Regions with zero occupancy should never be considered part of the experimental model, and consequently must be excluded from any interpretations. Occupancies higher than 1.0 result from obvious errors. When scrolling through a source PDB file, it may be useful to see if there were any alert flags set by the annotator, and, for the more inquisitive reader, to see what data quality is reported in the experimental section.

In addition to providing the above criteria, a respectable crystallographic publication should show the electron-density map on which the key conclusions hinge. The reader should be able to assess its quality, especially with reference to the contour level at which it is presented.

As discussed throughout this review, if both the coordinates and structure factors are available in the PDB, it is possible to independently assess the quality of published crystal structures and thus adjust expectations about the level of detail that may be safely accepted by the readers. Although some large-scale independent refinement efforts are under way, in which many deposited structures are re-refined using consistent protocols, in a vast majority of cases the readers will not be expected to repeat structure refinement and map analysis themselves.

It is very important to apply some common-sense tests before taking structural results as an absolute proof of the biological properties of macromolecules. Does the proposed active site and the mechanism of action make sense? Clearly, the ‘active site’ of frankensteainase, with only hydrophobic residues present (Fig. 1A), is unlikely to be able to catalyze any known chemistry. If metal ions are important for structure interpretation, have they been correctly assigned? Again, the example of frankensteainase, in which a putative Mg²⁺ ion does not have the expected coordination (Fig. 1B), shows that strong proofs of the metal identity should be present.

Although we have provided a number of examples showing that not all published structures yield the same level of information, we should stress again that, by and large, an overwhelming majority of crystal structures can be safely assumed to be unquestionably correct. It is important to keep in mind that crystallography is the only method that has extensive built-in quality control criteria of the structural ‘product’. In electron microscopy, the model does have a definitive resolution, but typically it is at least an order of magnitude worse than in X-ray crystallography. NMR-derived models do not possess any resolution and their quality cannot be assessed by reference to experiment using a criterion such as the R-factor. They can be evaluated, however, by similar rmsd(bond) and Ramachandran plot criteria. Finally, although we gave some general guidelines for the interpretation of the indicators of structure quality, we must stress that there is some level of subjectivity in their interpretation, and that other crystallographers may not exactly agree with all of our recommendations. That, however, is the beauty of the crystallographic method it is always open to further ‘refinement’.

Acknowledgements

We would like to thank Heping Zheng for helping with identification of the PDB files mentioned in this review. Original work in the laboratories of AW and ZD was supported by the Intramural Research Program of the NIH, National Cancer Institute, Center for Cancer Research, and WM was supported by grant GM74942 and GM53163. The research of MJ was supported by a Faculty Scholar fellowship from the Center for Cancer Research of the National Cancer Institute.

References

A. Wlodawer et al.

Protein crystallography for non-crystallographers


